

*Application for*  
**UNITED STATES LETTERS PATENT**

*Of*

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*For*

**METHOD FOR PRODUCING NANOPARTICLE LAYER HAVING UNIFORM EASY  
AXIS OF MAGNETIZATION, MAGNETIC RECORDING MEDIUM HAVING SUCH  
LAYER, ITS PRODUCTION METHOD, AND ITS PRODUCTION APPARATUS**

TITLE OF THE INVENTION

METHOD FOR PRODUCING NANOPARTICLE LAYER HAVING  
UNIFORM EASY AXIS OF MAGNETIZATION, MAGNETIC RECORDING  
5 MEDIUM HAVING SUCH LAYER, ITS PRODUCTION METHOD, AND ITS  
PRODUCTION APPARATUS

PRIORITY CLAIM

This application claims priority under 35 U.S.C. §119  
10 to Japanese patent application P2003-005242 filed January  
14, 2003, the entire disclosure of which is hereby  
incorporated herein by reference.

15 FIELD OF THE INVENTION

This invention relates to a magnetic, a  
thermomagnetic, or a magnetooptical recording medium used  
in a magnetic disk system or the like, a method for recording  
using such magnetic recording medium, and an apparatus for  
20 producing such magnetic recording medium.

BACKGROUND OF THE INVENTION

With the recent increase in the capacity of the  
magnetic recording system, attempts have been made to  
25 increase recording density of the magnetic recording medium.

In order to increase the density of the recording bit on the magnetic recording medium, decrease in the noise of the medium is necessary, and for this, use of smaller magnetization reversal units on the magnetic recording 5 layer is required. Reduction in the size of the magnetic crystal grains constituting the magnetic recording layer has been found effective for such increase in the recording density. However, use of excessively minute magnetic crystal grains is known to invite thermal demagnetization 10 wherein magnetization on the magnetic recording layer becomes thermally unstable. Use of magnetic crystal grains having a uniform size distribution is important to reduce the thermal demagnetization. In other words, size reduction of the magnetic crystal grains simultaneously with the 15 reduction in the grain size dispersion or standard deviation is required in the medium adapted for use in high density recording.

Conventional magnetic recording mediums have been produced by sputtering a seed layer, an underlying layer, 20 a magnetic layer functioning as a recording layer, a protective layer, and the like in this order on a circular glass or aluminum substrate. In the magnetic layer formed by sputtering, size dispersion of the magnetic crystal grains constituting the magnetic layer is large. The size 25 dispersion and the average grain size, however, can be

reduced in the case of sputtering by controlling the conditions of the film deposition. Still, the control of the grain size dispersion is difficult, and it is said that the grain size dispersion is limited to the level of about 5 20%.

An attempt to overcome the need for reducing the size and size dispersion degree of the magnetic crystal grains is disclosed in Patent Document 1, (Japanese Patent Laid-Open No. 2000-48340, corresponding to US patent 10 6,162,532) and a document relevant to this Patent Document 1, is Non-Patent Document 1, Science, vol. 287, pages 1989 to 1992 (issue of March 17, 2000).

In Patent Document 1 and Non-Patent Document 1, the magnetic nanoparticles constituting the recording layer are 15 produced not by the conventional sputtering but by a chemical synthesis. In Non-Patent Document 1, FePt alloy (uniaxial anisotropy constant, Ku:  $7 \times 10^6$  J/m<sup>3</sup>) which is a hopeful candidate for the near future high recording density is synthesized in an organic solvent by reacting an 20 iron pentacarbonyl compound (Fe(CO)<sub>5</sub>) and an acetylacetone platinum compound (Pt(acac)<sub>2</sub>). According to the Patent Document 1 and the Non-Patent Document 1, magnetic nanoparticles having an arbitrary diameter in the range of at least 3 nm and up to 10 nm with the size dispersion

standard deviation of 5 to 10% could be selectively produced by using the chemical synthesis as described above.

The magnetic nanoparticle produced by the chemical synthesis as described in the Patent Document 1 and the 5 Non-Patent Document 1 comprises a magnetic metal as indicated 1 in FIG. 1, which comprises either a single magnetic metal element or an alloy containing at least one magnetic metal element. Such magnetic nanoparticle is coated with an organic compound as indicated by 2. This 10 coating of the organic compound improves adhesion both between the magnetic nanoparticles and the substrate surface and between the adjacent magnetic nanoparticles, and there is disclosed that such organic compound coating facilitates the stable production of the ordered array of 15 the magnetic nanoparticles in the formation of the monolayer or multilayer film. FIG. 2 shows a monolayer film of magnetic nanoparticles. In FIG. 2(a), the layer of magnetic nanoparticle layer 5 is formed on the underlying layer or the soft magnetic layer 4 formed on the substrate 3, and the 20 magnetic nanoparticle 1 is covered with the coating 2.

In addition to the role as described above, the coating of the organic compound is believed to play an important role of improving the storage stability of the colloid solution of the magnetic nanoparticles. The 25 presence of the organic compound coating between the

magnetic nanoparticles in the resulting film is also believed to reduce the magnetic interaction between the adjacent magnetic nanoparticles. This phenomenon may be similar to the phenomenon found in the medium having the 5 layer of CoCrPt, CoCrTa, or the like formed by sputtering wherein Cr segregated layer is formed at the boundary of the magnetic crystal grains.

Typical organic compounds used for the coating in the Patent Document 1 are organic materials containing a long 10 chain organic compound represented by the formula: R-X wherein R is desirably a member selected from straight and branched hydrocarbon and fluorocarbon chains containing 6 to 22 carbon atoms, and X is desirably a member selected from carboxylic acids, phosphonic acids, phosphinic acids, 15 sulfonic acids, sulfinic acids, and thiols, among which oleic acid being mentioned as the most desirable for use as the coating.

Non-Patent Document 1 describes that, when the recording layer comprising magnetic nanoparticles formed 20 was subjected to a high temperature heat treatment at about 560°C, the coating of the organic compound such as oleic acid did not evaporate, but became carbonized as indicated by 6 in FIG. 2(b) and remained around the magnetic nanoparticles. Such carbonized organic substance remaining between the 25 magnetic nanoparticles is believed to contribute for the

reduction of the magnetic interaction between the magnetic particles. Non-Patent Document 1 also describes that crystallographic structure of the FePt magnetic nanoparticles changes by the heat treatment from the fcc structure at the time of its chemical sysnthesis into the ordered structure L10. In the case of FePt, magnetism is not found in the fcc structure, and ferromagnetism is developed when it takes the ordered structure. It is to be noted that the magnetic field was not applied in the heat treatment after the film formation. Accordingly, the easy axis of magnetization of the magnetic nanoparticles is believed to be randomly oriented.

In the technology described in Non-Patent Document 1, the nanoparticle layer formed is subjected to a high temperature treatment at about 500°C to 600°C to thereby convert the nanoparticle crystal structure from fcc structure to L10 ordered structure to thereby magnetize the nanoparticles to the degree sufficient for use as a recording medium. As a result of such high temperature heat treatment, the nanoparticle layer experiences disturbance in the array of the nanoparticles as well as agglomeration of the nanoparticles, and when such nanoparticle layer is used in a magnetic recording layer, the layer suffers from an insufficient flatness. The high temperature heat treatment also results in the undesirable deterioration of

the underlying layer, the soft magnetic layer, and the like between the nanoparticle layer and the substrate. In spite of the high magnetization degree of the nanoparticle layer after the high temperature heat treatment, it is difficult 5 to use such nanoparticle layer in a magnetic recording medium wherein the substrate is actually rotated for the reading and writing of the information by the read head.

On the other hand, in the technology described in Patent Document 1, the easy axis of magnetization of the 10 magnetic nanoparticles constituting the recording layer is randomly oriented, and orientation of the easy axis of magnetization in a particular direction such as in-plane direction of the medium or thickness direction of the medium is difficult. As a consequence of such difficulty, the 15 resulting magnetic recording layer suffers from inferior magnetic properties compared to the conventional in-plane recording or perpendicular recording medium.

## 20 SUMMARY OF THE INVENTION

In view of the situation as described above, the present invention may include providing a magnetic recording medium having a nanoparticle layer wherein the high temperature heat treatment that had been conducted for 25 magnetization of the nanoparticles is no longer necessary,

flatness of nanoparticle layer has been improved; the underlying layer and the soft magnetic layer do not experience deterioration, easy axis of magnetization of the nanoparticles is substantially parallel to a direction 5 which is at a particular angle to said substrate surface, and excellent magnetic properties are realized. Other features of the invention may include to providing a method for producing such medium and apparatus used in producing such medium.

10 The features as described above are attained by using a magnetic recording medium at least comprising a substrate having a surface; and a nanoparticle layer comprising an array of nanoparticles having an average particle size of at least 1 nm and not more than 20 nm, and containing at least 15 one element selected from Fe, Co, Ni, Mn, Sm, Pt, and Pd, and an organic compound between said array of nanoparticles; wherein easy axis of magnetization of said nanoparticles is substantially parallel to a direction which is at a particular angle to said substrate surface. Such magnetic recording medium can be produced by a method for producing 20 a magnetic recording medium comprising the steps of: forming a nanoparticle layer on a substrate having a surface or on an underlying layer or a soft magnetic layer formed on said substrate by arranging particles in a substantially ordered 25 array, said particles each comprising a nanoparticle and an

organic compound coating said nanoparticle, and said nanoparticles having an average particle size of at least 1 nm and not more than 20 nm, and containing at least one element selected from Fe, Co, Ni, Mn, Sm, Pt, and Pd;

5 irradiating said nanoparticle layer with an infrared beam to magnetize said nanoparticles and produce magnetic nanoparticles; applying a magnetic field to said nanoparticle layer to orient easy axis of magnetization of said magnetic nanoparticles in a substantially uniform

10 direction; and irradiating said nanoparticle layer with an ultraviolet beam to bind said organic compound. In addition, such magnetic recording medium can be produce by an apparatus having an infrared irradiating section for irradiating a particular region of the substrate having the

15 nanoparticle layer formed thereon with an infrared beam; a magnetic field applying section for applying a magnetic field to said particular region after the irradiation of the infrared beam; and an ultraviolet irradiating section for irradiating said particular region with an ultraviolet beam

20 after the application of the magnetic field.

#### BRIEF DESCRIPTION OF THE DRAWING

FIG. 1 is a view showing prior art nanoparticles covered with a coating;

FIGS. 2A-2B are prior art cross-sectional views of the magnetic recording medium having a nanoparticle layer;

Fig. 3A-3B are views showing an apparatus for producing the magnetic recording medium having a nanoparticle layer wherein the easy axis of magnetization of nanoparticles is oriented in the same direction parallel to the substrate.

Figs. 4A-4B are views showing an apparatus for producing the magnetic recording medium having a nanoparticle layer wherein the easy axis of magnetization of nanoparticles is oriented in the same direction inclined to the substrate at a 45-degrees.

Fig. 5A-5B is a view showing an apparatus for producing the magnetic recording medium having a nanoparticle layer wherein easy axis of magnetization of nanoparticles is oriented in the same direction vertical to the substrate.

Fig. 6 is a side view of the magnetic recording medium having a nanoparticle layer wherein easy axis of magnetization of nanoparticles is oriented in the same direction vertical to the substrate.

Figs. 7A - 7D are side views showing a manufacturing process for producing the magnetic recording medium having a nanoparticle layer wherein the easy axis of magnetization

of nanoparticles is oriented in the same direction vertical to the substrate.

Figs. 8A - 8C are top and side views showing a manufacturing process for producing a magnetic recording medium having a nanoparticle layer by Langmuir-Blodgett method.

Fig. 9A - 9C are side views showing a prior art manufacturing process for producing a magnetic recording medium having a nanoparticle layer by spin coating method.

Fig. 10A - 10B are a side view and a perspective view showing a magnetic read/write processes by using a head system comprising separate read and write heads.

Fig. 11A - 11B are a side view and a perspective view showing a optically assisted magnetic read/write processes by using a head system comprising separate read and write heads.

#### DETAILED DESCRIPTION OF THE PREFERRED EMBODIMENTS

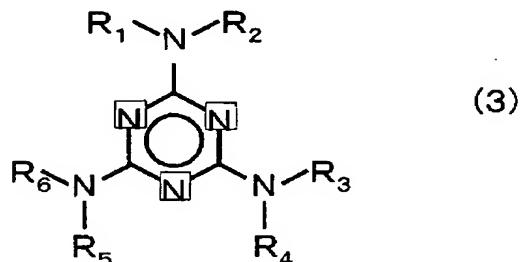
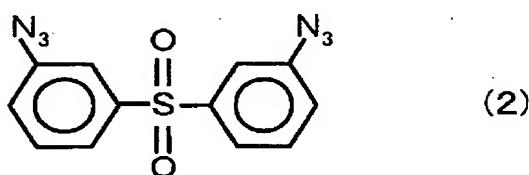
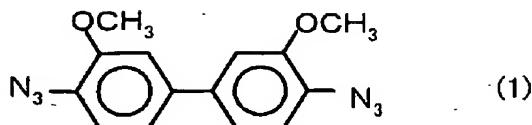
In the magnetic recording medium as described above, the nanoparticles may contain at least one magnetic metal element selected from Fe, Co, Ni, Mn, Sm, Pt, Pd, and the like. The nanoparticles may also be magnetic nanoparticles comprising an intermetallic compound of the aforesaid elements, a binary alloy of said elements, or a ternary alloy

of said elements. In view of the expected higher recording density in near future, the preferred are magnetic nanoparticles having the composition of FePt or FePd having a large uniaxial anisotropy constant (Ku), or a ternary 5 alloy comprising FePt or FePd and a third element. The third element used may be Cu, Ag, Au, Ru, Rh, Ir, Pb, or Bi, as well as other elements. Magnetic nanoparticles having a structure comprising the core of a binary alloy which is typically FePt or FePd and the surrounding shell comprising 10 the aforementioned ternary element, Pt or Pd are also useful.

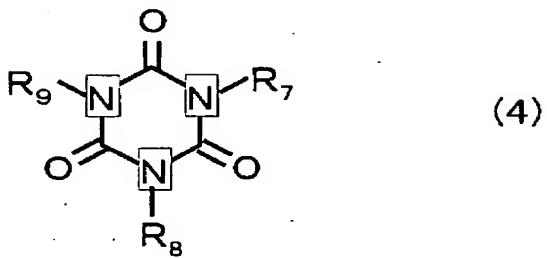
The organic compound which is present between the array of nanoparticles may be the organic compound coating the nanoparticles. Such organic compound may be an 15 unsaturated fatty acid compound such as oleic acid, or an amine compound of an unsaturated fatty acid such as oleylamine. The compounds which may be used also include a compound having thiol group, as well as a compound having at least one carbon-carbon double bond or triple bond. Other 20 organic compounds may also be used for such coating.

The organic compound between the array of nanoparticles may further contain a compound which is capable of binding the organic compound coating the nanoparticles when it is irradiated with a light beam or a 25 radiation or by applying heat. To be more specific, the

compound represented by the following general formulae (1) to (4) may be used.



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In the formulae, R<sub>1</sub> to R<sub>9</sub> are independently a functional group selected from carboxylic acids, phosphonic acids, phosphinic acids, sulfonic acids, sulfinic acids, 10 thiols, hydroxyls, and hydrogen atom; or a hydrocarbon group containing carbon-carbon double bond or ether bond. The compound represented by the general formulae (1) to (4) are

crosslinking agents, and crosslinking agents having other structures may also be used.

The recording layer of the magnetic recording medium according to the present invention may be constituted from 5 a monolayer film or a multilayer film of nanoparticles. The monolayer or the multilayer nanoparticle layer may be formed by using Langmuir-Blodgett (LB) method as shown in FIG. 8A - 8C. When LB method is used, the nanoparticle layer may be formed by the procedure as described below. First, a 10 colloid solution of the nanoparticles which have been coated with an organic compound 2 is gradually added dropwise onto the surface of clean water 22 that has been filled in a trough 21 to form a monolayer film of nanoparticles wherein the nanoparticles are sparsely arranged. Next, after 15 evaporating the solvent, the nanoparticle monolayer film of sparsely arranged nanoparticles is gently compressed by a moving barrier 23 to the direction 24, and when the compression is terminated at the pressure wherein the distance between the nanoparticle is at its least while 20 maintaining the form of the monolayer film, the film wherein the nanoparticles are packed at their closest is obtained. Then, the substrate or the substrate having an underlying layer or a soft magnetic layer formed thereon held at horizontal position is brought in contact with the water 25 surface, and pulled up to thereby transfer the monolayer

film onto the substrate and obtain the Langmuir-Blodgett (LB) film comprising the monolayer film of nanoparticles. A LB multilayer film comprising a laminate of nanoparticle monolayer films may also be produced by repeating the 5 procedures as described above.

The recording layer comprising the nanoparticle layer may also be formed by spin coating as shown in FIG. 9A - 9C wherein the colloid solution 25 of the nanoparticles is dropped onto the surface of the substrate and a thin film 10 is formed by rotating the substrate to the direction 26. When the molecular weight and the molecular structure of the compound coating the nanoparticles is adequately selected, and the concentration of the colloid solution is adjusted, and the rotation conditions are optimized, production of a 15 recording layer comprising a substantially ordered array of closely packed nanoparticles is enabled. Methods other than those described above may also be employed for producing the recording layer comprising the nanoparticle layer.

The nanoparticles in the thus formed nanoparticle 20 layer have cubic crystal fcc structure, and the nanoparticles are scarcely magnetized. Therefore, crystallographic structure of the nanoparticles needs to be converted to L10 ordered structure for magnetization. Referring to Figures 7A -7D, when the nanoparticle layer is 25 irradiated with an infrared beam 9, the infrared beam is

absorbed by the nanoparticles comprising a metal element 1 and turns into heat which causes partial change in crystallographic structure of the nanoparticles. The infrared beam 9 is well absorbed by the nanoparticles 5 comprising a metal element 1 while it is less likely to be absorbed by the organic compound 2 or the crosslinking agent coating the nanoparticles, and therefore, the crystallographic structure of the nanoparticles can be converted from the cubic crystal fcc to L10 ordered 10 structure for magnetization 20 of the nanoparticles without changing the quality of the organic compound 2 between the nanoparticles by adjusting the intensity and irradiation time of the infrared beam. Degree of the conversion of the nanoparticles from the cubic crystal fcc to the L10 ordered 15 structure can be controlled by means of the infrared beam irradiated in this procedure. Conversion to the ordered structure can be proceeded to the level of 100% for further magnetization and ferromagnetism by increasing the intensity or the irradiation time of the infrared beam. The 20 infrared beam used may preferably have a long wavelength of 600 nm or longer, and an infrared laser beam may be used for the infrared beam.

After the magnetization as described above, as shown in FIG. 7C, a magnetic field 16 is applied to orient the easy 25 axis of magnetization of the each nanoparticle to a

direction substantially parallel to a direction at a particular angle with the substrate surface. In this process, the direction of the magnetic field may be set parallel to the substrate surface, at 45 degrees to the 5 substrate surface, perpendicular to the substrate surface, or at another selected angle. The magnetic field used may be either a static magnetic field wherein the direction and the intensity of the magnetic field does not change with time, or a pulse magnetic field wherein the direction of the 10 magnetic field is constant while the intensity of the magnetic field alters with time. When the magnetic field is applied to the nanoparticle layer which has been magnetized as described above, the easy axis of magnetization of the each nanoparticle can be oriented to 15 a direction substantially parallel to a direction at a particular angle with the substrate surface.

Next, as shown in FIG.7D, the nanoparticle layer is irradiated with an ultraviolet beam 13 to fix the direction of the easy axis of magnetization. The ultraviolet beam 20 irradiated is absorbed by the organic compound 2 between the nanoparticles to induce photochemical or thermochemical reaction to crosslink or bind the organic compound. When an organic compound such as a crosslinking agent capable of binding the organic compound coating the nanoparticles is 25 present between the nanoparticles, crosslinking efficiency

will be improved. The ultraviolet beam irradiated is preferably a short wavelength beam having a wavelength of up to 400 nm. The crosslinking efficiency can be further improved by adequately adjusting the structure of the 5 crosslinking agent for crosslinking the organic compound between the nanoparticles, and also, by adjusting the wavelength, the intensity, and the irradiation time of the ultraviolet beam.

When the magnetization by the irradiation of the 10 infrared beam is insufficient, the nanoparticles may be further ordered by conducting a heat treatment at a temperature of up to 300°C for an arbitrary period after the step of binding the organic compound by the ultraviolet irradiation.

15 When a nanoparticle layer is irradiated with an infrared beam for magnetization, a magnetic field is then applied at a particular angle with the substrate surface to orient the easy axis of magnetization of the nanoparticles in the direction of the magnetic field, and the nanoparticle 20 layer is further irradiated with an ultraviolet beam to crosslink the organic compound between the nanoparticle to thereby fix the nanoparticles as described above, a nanoparticle layer wherein the easy axis of magnetization is substantially parallel to a direction at a particular 25 angle to the substrate surface can be obtained. In this

procedure, when a magnetic field perpendicular to the substrate surface is applied, while adequately adjusting the intensity and the time of the magnetic field application, the layer obtained will be a perpendicular magnetic layer 5 wherein number of nanoparticles wherein angle between the direction perpendicular to the substrate surface and the easy axis of magnetization of the nanoparticles is up to 5 degrees is at least 90% of the total number of nanoparticles included in the nanoparticle layer. Such magnetic layer 10 exhibits favorable perpendicular magnetic anisotropy as well as excellent magnetic properties.

The recording of the information on the nanoparticle medium having the nanoparticle layer exhibiting the favorable perpendicular magnetic anisotropy as described 15 above may be accomplished by a perpendicular magnetic recording system wherein the main component of the leakage magnetic field from the write head is perpendicular to the in-plane direction of the substrate. The recording may be also accomplished by a thermomagnetic or a magneto-optical 20 recording system wherein magnetic recording is conducted while the recording area of the medium is selectively irradiated with heat or light.

The apparatus used for producing a magnetic recording medium wherein easy axis of magnetization of the 25 nanoparticles is oriented at a direction which is at a

particular angle to the substrate surface may be the apparatus as shown in FIG. 3. This apparatus has a rotating section 8 which rotates the substrate 3 (see rotation direction 17 indicated) bearing the nanoparticle layer 5 at 5 an arbitrary rotation speed around a particular rotation axis 7, and in this apparatus, an infrared irradiating section 10 for irradiating an infrared beam 9 to a particular region of said substrate, a magnetic field applying section 12 having coils 11 for applying a magnetic field to said 10 particular region after the infrared irradiation (see magnetic field direction 16), and an ultraviolet irradiating section 14 for irradiating a ultraviolet beam 13 to said particular region after the magnetic field application are concentrically arranged around on a circle 15 15 whose center is the rotation center. After such procedure, the organic compound coating the nanoparticle 1 or the compound 6 derived from the organic compound coating the nanoparticle will be present between the nanoparticles.

20 EXAMPLES

Next, the present invention is described in further detail by referring to the following Examples which by no means limit the scope of the invention.

Example 1

Referring to Figures 5A & 5B, Spherical magnetic nanoparticles having a particle dispersion standard deviation degree of up to 10% and a diameter in the range of 1 to 20 nm were chemically synthesized, and the 5 nanoparticles were classified by size in a centrifuge such that each class had a diameter dispersion standard deviation of up to 5%. In the thus produced colloid solution of the nanoparticles, nanoparticles comprising a magnetic metal element surrounded by a coating of an organic compound were 10 dispersed as colloid. Next, the colloid solution of the nanoparticles as described above was dropped onto a soft magnetic layer which had been deposited on a glass substrate by sputtering, and the substrate was rotated for spin coating of the colloid solution to obtain a monolayer film 15 of the nanoparticles which was then subjected to a prebaking at 80°C for 5 minutes. The substrate having the thus formed nanoparticle layer thereon was rotated such that an arbitrary region of the nanoparticle layer was irradiated with an infrared beam 9 having a wavelength of 800 nm, and 20 a magnetic field 16 in the direction perpendicular to the substrate surface was applied to this region at the very moment when this region passed between a pair of coils 11 having magnetic poles arranged on opposite sides of the substrate, and this region was further irradiated with an 25 ultraviolet beam 13 having a wavelength of 200 nm

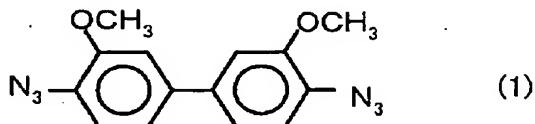
immediately after passing between the coils. A perpendicular magnetic recording medium having a nanoparticle layer wherein the easy axis of magnetization 18 as shown in FIG. 6 is oriented at a direction 19 5 perpendicular to the in-plane direction of the substrate is thereby produced. This is the best mode.

Example 2

A monolayer film of the nanoparticle was formed by Langmuir-Blodgett method as shown in FIG. 8A - 8C instead of 10 the spin coating used in Example 1. The substrate having the thus formed nanoparticle layer thereon was rotated such that an arbitrary region of the nanoparticle layer was irradiated with an infrared beam having a wavelength of 800 nm, and a magnetic field in the direction perpendicular to 15 the substrate surface was applied to this region at the very moment when this region passed between a pair of coils having magnetic poles arranged on opposite sides of the substrate, and this region was further irradiated with an ultraviolet beam having a wavelength of 200 nm immediately after passing 20 between the coils. A perpendicular magnetic recording medium having a nanoparticle layer wherein the easy axis of magnetization had been oriented at a direction perpendicular to the in-plane direction of the substrate was thereby produced.

Example 3

To the colloid solution of nanoparticles used in Example 1 was added a crosslinking agent represented by the general formula (1):



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at an amount of 20% by weight of the colloid. The colloid solution having the crosslinking agent added thereto was dropped onto the surface of clean water to form a LB monolayer film of nanoparticles by Langmuir-Blodgett method.

10 This LB monolayer film was transferred onto the substrate. When this LB monolayer film was observed under SEM, the resulting array of nanoparticles substantially had closest packed structure.

The substrate having the thus formed nanoparticle  
15 layer thereon was rotated such that an arbitrary region of the nanoparticle layer was irradiated with an infrared beam having a wavelength of 800 nm, and a magnetic field in the direction perpendicular to the substrate surface was applied to this region at the very moment when this region  
20 passed between a pair of coils having magnetic poles arranged on opposite sides of the substrate, and this region was further irradiated with an ultraviolet beam having a wavelength of 200 nm immediately after passing between the

coils. A perpendicular magnetic recording medium having a nanoparticle layer wherein the easy axis of magnetization had been oriented at a direction perpendicular to the in-plane direction of the substrate was thereby produced.

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Example 4

Referring to Figs. 4A and 4B, the procedure of Example 3 was repeated by using a colloid solution of nanoparticles having a crosslinking agent added thereto to form a nanoparticle monolayer film by the LB method. The substrate having the thus formed nanoparticle layer thereon was rotated such that an arbitrary region of the nanoparticle layer was irradiated with an infrared beam having a wavelength of 800 nm and a magnetic field at an angle of 45 degrees to the substrate surface was applied to this region at the very moment when the region passed between a pair of coils having a pair of magnetic poles arranged on opposite sides of the substrate at an angle of 45 degrees with the substrate, and this region was irradiated with an ultraviolet beam having a wavelength of 200 nm immediately after passing between the coils. A magnetic recording medium having a nanoparticle layer wherein the easy axis of magnetization had been oriented at 45 degrees to the in-plane direction of the substrate was thereby produced.

25 Example 5

Referring to Figure 5A and 5B, the magnetic nanoparticle medium produced in Example 3 was evaluated by using a sample vibration magnetometer. A magnetization curve exhibiting excellent magnetic properties including a 5 perpendicular coercive force of 800 kA/m (10000 Oe), a coercive force squareness ratio  $S^*$  of 0.8, and a residual magnetization of 200 emu/cc was obtained.

Example 6

Referring to Fig. 10A - 10B, the magnetic nanoparticle medium 27 produced in Example 3 was combined with a head system 36 comprising separate read and write heads employing a thin film single pole head for perpendicular magnetic recording for the write head 35 composed of an auxiliary pole 32, a main pole 33, and coils 34, and a GMR element 30 between 15 shields 28, 29 for the read head 31 to evaluate the output. A magnetic flux 38 orients the direction of magnetization 20 of the medium to the same direction to the magnetic field while the medium moves to the direction 37. At the output evaluation, a magnetic nanoparticle medium 27 is rotated to 20 the direction 40, and the head system 36 is mounted on an arm 39. A peak-to-peak output of about 1 mV was measured at a recording density of 100 kfci. The medium also exhibited an abrasion resistance equivalent to a conventional medium wherein the recording layer had been 25 formed by sputtering.

Example 7

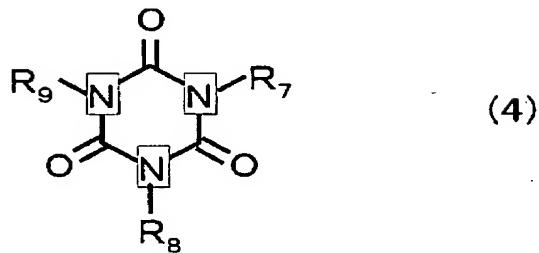
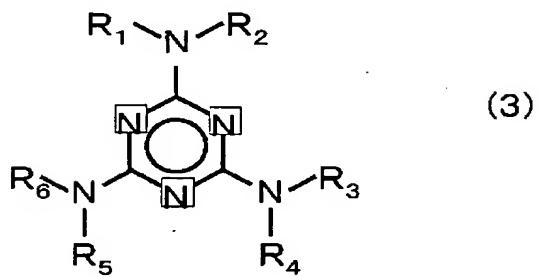
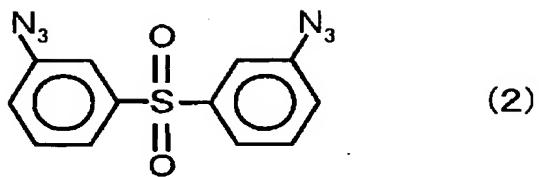
Referring to FIG.11A to 11B, a read/write experiment was conducted by using an optically assisted magnetic recording head 43 wherein only the recording area is heated by a light beam 42 from a laser 41 for the writing, and a GMR element 30 for the read head instead of the perpendicular magnetic recording used in Example 6. A peak-to-peak output of about 1 mV was measured at a recording density of 100 kfci.

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Example 8

The magnetic nanoparticle medium produced in Example 3 was observed under SEM. No disturbance in the particle array or agglomeration of the particles induced by the heat treatment were observed. Observation under AFM revealed that the medium had a surface roughness Ra of up to 0.8.

In Example 3 as described above, the crosslinking agent used was the one represented by the general formula (1). The crosslinking agent of the formula (1), however, could be replaced with a compound represented by any one of the general formulae (2), (3), and (4). It is of course possible to conduct the following Examples 4 to 8 by using the product prepared by using the compound represented by any one of the general formulae (2), (3), and (4).



In the formulae, R1 to R9 are independently a

5 functional group selected from carboxylic acids, phosphonic acids, phosphinic acids, sulfonic acids, sulfinic acids, thiols, hydroxyls, and hydrogen atom; or a hydrocarbon group containing carbon-carbon double bond or ether bond. R1 to R9 may be the same or different.

10 As described above, a magnetic recording medium provided with a magnetic recording layer wherein nanoparticles are arranged on the substrate in an ordered array and an organic compound is present between the nanoparticles could be produced, and in this medium, no high

temperature heat treatment was necessary, flatness of the nanoparticle layer was improved compared to the conventional medium comprising the magnetic nanoparticles, the underlying layer or the soft magnetic layer did not 5 experience deterioration, easy axis of magnetization of the nanoparticles was substantially parallel to a direction at a particular angle to said substrate surface, and magnetic properties were excellent. This medium could be recorded with information by magnetic recording.

10 Example 9

Referring to Figure 3A and 3B, the procedure of Example 3 was repeated by using a colloid solution of nanoparticles having a crosslinking agent added thereto to form a nanoparticle monolayer film by the LB method. The 15 substrate having the thus formed nanoparticle layer thereon was rotated such that an arbitrary region of the nanoparticle layer was irradiated with an infrared beam having a wavelength of 800 nm and a magnetic field in the direction parallel to the substrate surface was applied to this region at the very moment when the region passed the area in which a pair of coils having a pair of magnetic poles arranged on the same side of the substrate in the direction parallel to the substrate, and this region was irradiated with an ultraviolet beam having a wavelength of 200 nm 20 immediately after passing between the coils. A magnetic 25

recording medium having a nanoparticle layer wherein the easy axis of magnetization had been oriented in the direction parallel to the substrate was thereby produced.

Although the above examples are provided applicants 5 also envision variations and equivalents to the disclosure discussed above to be within the scope of this disclosure and the claims.